



# Synthesis of high dispersed intermetallic Ag<sub>4</sub>Sn/C and its enhanced oxygen reduction reaction activity

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## HIGHLIGHTS

- We have firstly synthesized carbon support nanoscale Ag<sub>4</sub>Sn intermetallic compound (Ag<sub>4</sub>Sn/C).
- The electrocatalytic performance of Ag<sub>4</sub>Sn/C towards oxygen reduction reaction is superior to Ag/C.
- Ag<sub>4</sub>Sn/C shows methanol tolerance property and good stability under DMAFC condition.

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## ABSTRACT

Ag<sub>4</sub>Sn/C catalyst is synthesized by a solution phase reduction method assisted with ultrasonic at 0 °C. The as-prepared catalyst is characterized by transmission electron microscopy (TEM), X-ray diffraction (XRD) and electrochemical measurements. The Ag<sub>4</sub>Sn intermetallic electrocatalyst displays enhanced electrocatalytic activity towards oxygen reduction reaction (ORR) and good methanol-tolerant property in alkaline media. The ORR catalyzed by Ag<sub>4</sub>Sn/C proceeds a four-electron pathway. The obtained results are important for the development of Ag-based catalysts in alkaline fuel cells.

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## 1. Introduction

Direct methanol alkaline fuel cell (DMAFC) has attracted much attention in recent years due to low operating temperature, less corrosive environment, and more facile kinetics for oxygen reduction [1,2]. It has great application potential in electrical vehicles, portable energy supplies, power station and aerospace, etc. However, the large scale commercialization of DMAFC technology is still hindered by the high cost, scarce resources and intolerance to methanol crossover of Pt-based cathode catalysts [3]. As a result, some alternative catalysts based on non-platinum metals have been explored, including the use of transition metal [4,5], carbon based materials [6], metal oxides [7], metal nitrides [8], and transition metal doped conducting polymer [9,10] in terms of their high conductivity, low price and favorable performances. Among those

options, the relatively inexpensive and abundant Ag is a top candidate for DMAFC. Yet the poor affinity between Ag and O<sub>2</sub>, which leads to difficulty in breaking O–O bond, is the main reason for the relatively low catalytic activity of Ag when compares to Pt [11]. Therefore, it is highly desirable to explore novel Ag-based catalyst for the development of DMAFC.

In previous works, the monometallic Ag electrocatalysts with small particle size and special morphologies exhibited better catalytic activity than bulk Ag in alkaline media [12,13]. AgPd alloy [14], Ag/Mn<sub>x</sub>O<sub>y</sub> [15] and AgCo bimetallic [11] showed enhanced ORR performance, which could be explained by the ensemble effect or the bifunctional mechanism. Although some impressive progresses have been brought about silver based materials, to the best of our knowledge, there haven't any reports about the electrocatalytic activity of silver based intermetallic nanoparticles towards ORR, as well as tolerance towards small organic molecules in fuel cell applications.

Pt-based intermetallics have been widely investigated as catalysts in fuel cell due to their unique ordered structures, like PtBi [16], PtFe [17], PtNi [18], etc. Chanho Pak et al. reported Pd<sub>3</sub>Sn

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intermetallic has enhanced catalytic performance compared to PdSn alloy [19]. The mechanism of enhanced catalytic performance could be attributed to the electronic interaction between Pd and Sn, which caused a combined effect of reducing oxygen binding energy of Pd, and increasing  $O_2$  adsorption affinity. Herein, from the point of view for catalyst design,  $Ag_4Sn$  could be a highly potential candidate as effective fuel cell catalyst relative to monometallic Ag. Previous works showed that  $Ag_4Sn$  was prepared through high temperature polyol process using poly(vinyl pyrrolidone) as capping agent. The as-prepared  $Ag_4Sn$  suffered from particles agglomeration [20] and block adsorption in active site by capping agent, which could suppress catalytic activity.

In this work, we reported a novel carbon supported  $Ag_4Sn$  intermetallic catalyst prepared by a solution phase co-reduction method assisted with ultrasonic at  $0^\circ C$  in the absence of capping agent. The as-synthesized  $Ag_4Sn$  particles with average size of approximate 5 nm are uniformly dispersed on carbon support. Compared to Ag/C,  $Ag_4Sn/C$  was shown to have superior ORR activity and enhanced stability properties. Furthermore, high methanol tolerance in alkaline solution also proved that  $Ag_4Sn/C$  is a good candidate catalyst for ORR in DMAFC application.

## 2. Experimental section

### 2.1. Preparation and characterization

Carbon supported  $Ag_4Sn$  nanoparticles were synthesized by one-pot reduction of Ag and Sn precursor in EG/isopropanol mixed solution in an ultrasonic cleaning bath (45 kHz). In a typical preparation process, 0.6 mmol stannous octanoate and 0.2 mmol silver nitrate ( $AgNO_3$ ) were dissolved into a mixed solution of 100 ml isopropanol and 200 ml ethylene glycol (EG). Then, 50 ml EG solution which contained 0.0783 g Vulcan XC-72 was gently added to the mixed solution under vigorous stirring. After purged with nitrogen for 1 h at  $0^\circ C$ , sodium borohydride ( $NaBH_4$ ) aqueous solution (0.2 M, 80 ml) was dropwisely added into the reaction solution, and kept stirring for another 6 h at  $0^\circ C$  under nitrogen atmosphere in an ultrasonic cleaning bath. The resulting solid was filtered and washed with copious ethanol for several times, then dried at  $80^\circ C$  for 12 h. In order to evaluate the influence of reaction temperature on product composition, control sample was also synthesized by the same procedure at room temperature. And for comparison use in electrochemical measurement, Ag/C and Sn/C was also synthesized at  $0^\circ C$ .

Crystal structure of the products was characterized by X-ray diffraction (XRD, Bruker D8 Advance,  $Cu K\alpha$  radiation), and the morphology was investigated by transmission electron microscopy (TEM, Tecnai F30). Surface status was determined by using XPS (AXIS-Ultra instrument from Kratos Analytical, Al  $K\alpha$  radiation,  $h\nu = 1486.6$  eV). To compensate for the effects of surface charges, the binding energies were calibrated using the C 1s hydrocarbon peak at 284.80 eV. The amount of Ag and Sn were determined by inductively coupled plasma-atomic emission spectroscopy (ICP-AES, PROFILE SPEC, Leeman) and TG analysis (EXSTAR TG/DTA 7300). During thermogravimetric analysis, the samples were heated from room temperature to  $600^\circ C$  at  $5^\circ C min^{-1}$ , and the air flow was precisely controlled at  $200 mL_{air} min^{-1}$ .

### 2.2. Electrochemical measurements

Electrochemical measurements were carried out by SP-240 potentiostat (Bio-Logic SAS, France). The glassy carbon rotating disk electrode (GC-RDE) modified with catalysts ink (geometrical area  $0.1257 cm^2$ ) was used as working electrode in a conventional three electrode cell. Hg/HgO and glassy carbon (GC) were used as reference and counter electrode, respectively. It should be noted that all

electrochemical measurements were further calibrated with respect to reversible hydrogen electrode (RHE) in order to give a more straightforward view of the catalysts performance in alkaline fuel cell applications. Catalysts inks were prepared by a typical process. 4 mg catalysts were suspended 30 min in 1 ml isopropanol which contained 80  $\mu l$  0.5 wt% Nafion solutions. The 10  $\mu l$  of the suspension was pipetted onto the GC-RDE substrate and then dried in ambient atmosphere. Cyclic voltammetry (CV) measurements were carried out in  $N_2$  saturated 0.1 M KOH between 0.213 V and 1.513 V vs. RHE. Polarization tests were performed in 0.1 M KOH with constantly purged  $O_2$  at a scan rate of  $10 mV s^{-1}$  between 0.213 V and 0.913 V. Chronoamperometry and methanol-tolerant tests were conducted at 0.713 V in  $O_2$  saturated 0.1 M KOH solution. The electrochemical measurements were all conducted at room temperature, i.e.,  $25 \pm 1^\circ C$ .

## 3. Results and discussion

Fig. 1a shows the X-ray diffraction (XRD) patterns of as-prepared carbon supported  $Ag_4Sn$  intermetallic, in which characteristic peaks of chemically ordered hexagonal structure  $Ag_4Sn$  can be observed (JCPDS No. 29-1151), in addition to the minor Ag phase and bump corresponding to carbon support. While the XRD peaks of the control sample prepared under room temperature are in good agreement with fcc Ag (JCPDS No. 65-2871), as shown in Fig. 1b. Although the high concentration of stannous octanoate was added during the synthesis, no obvious peaks corresponding to tin-related compounds can be observed in the X-ray pattern. This phenomenon might be explained by the competitive more facile adsorption of Ag on the carbon surface sites, and the difficulty in reducing of stannous octanoate to metallic Sn at room temperature. This result is also consistent with Gyenge's [21].

Fig. 2 shows the TEM images of  $Ag_4Sn/C$  and Ag/C nanoparticles with average particle size of 5 nm and 8 nm, respectively, which are obviously smaller than many previous reported Ag-based materials [11,22]. This could be attributed to the slow mass transfer rate resulted from the high viscosity of EG at low temperature. As a result, high dispersion ultra-fine intermetallic  $Ag_4Sn$  particles on carbon support could be one step obtained. It has been shown that the catalyst with high surface area can enhance the activity towards oxygen reduction reaction. Therefore, this synthesis route presented in this paper is very promising to prepare high dispersion ultra-fine intermetallic particles.

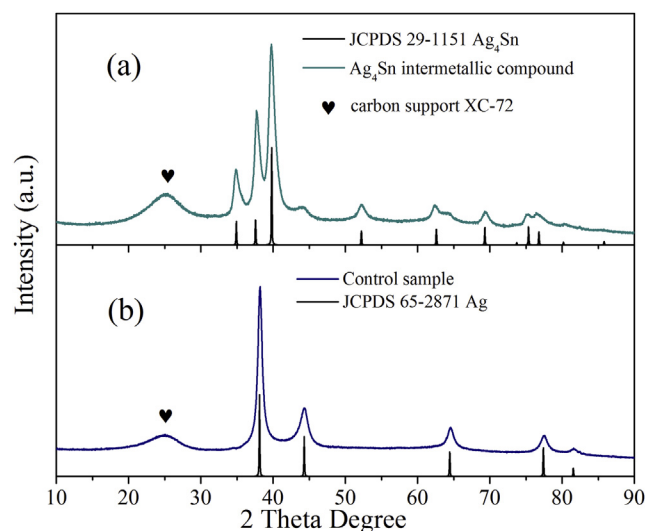


Fig. 1. XRD patterns of (a) carbon supported  $Ag_4Sn$  intermetallic compound prepared at  $0^\circ C$ , and (b) control sample prepared under room temperature.

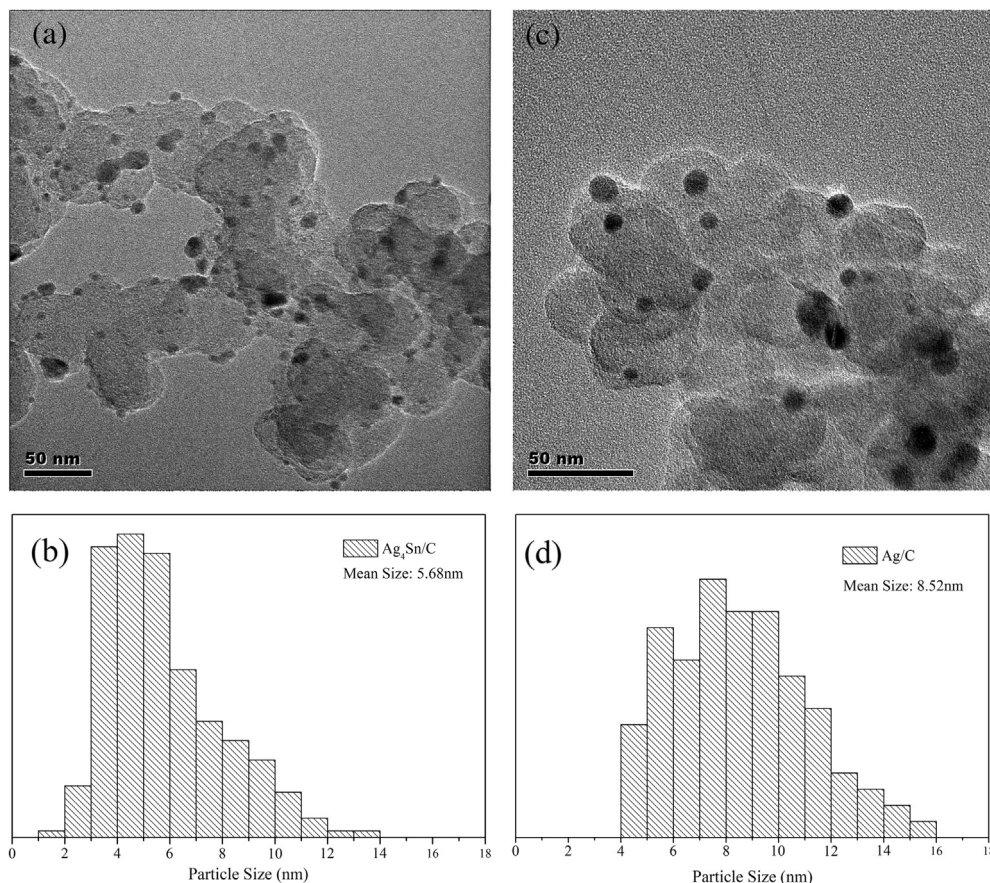


Fig. 2. TEM images and size distributions of  $\text{Ag}_4\text{Sn/C}$  (a), (b), and  $\text{Ag/C}$  (c), (d).

Chemical compositions of as-prepared catalysts were determined using inductively coupled plasma-atomic emission spectroscopy and TG analysis. Since  $\text{Ag}^+$  could so easily react with trace amount of  $\text{Cl}^-$  ion in ICP-AES system that results in the deviation of chemical composition analysis, thereby in this work, only weight percentage of Sn was determined through ICP-AES to be 7% in  $\text{Ag}_4\text{Sn/C}$ , 27% in  $\text{Sn/C}$  respectively. Since the residue of  $\text{Ag}_4\text{Sn/C}$  sample after TG analysis was Ag and  $\text{SnO}_2$  (determined by XRD, not shown here), therefore based on the following equation [23],  $\text{SnO}_2(\text{wt}\%) = \text{Sn}(\text{wt}\%) \times (\text{molecular weight of } \text{SnO}_2 / \text{molecular weight of Sn}) \times 100$ , the weight percentage of Ag in  $\text{Ag}_4\text{Sn/C}$  and  $\text{Ag/C}$  were determined to be 21% and 20%, respectively, according to TG analysis [14].

The XPS spectra of in  $\text{Ag/C}$  and  $\text{Ag}_4\text{Sn/C}$  are shown in Fig. 3. The doublets in Fig. 3a are assigned to Sn  $3d_{3/2}$  and  $3d_{5/2}$  state of either  $\text{SnO}_2$  or  $\text{SnO}$  at the surface of  $\text{Ag}_4\text{Sn/C}$ , since the binding energies of these two oxides are generally indistinguishable [21]. Fig. 3b presents XPS spectra of Ag  $3d$  orbits. Virtually there are no obvious distinction of binding energy for Ag  $3d$  orbits between  $\text{Ag/C}$  and  $\text{Ag}_4\text{Sn/C}$  (Ag  $3d_{5/2}$  368.65 eV and Ag  $3d_{3/2}$  374.65 eV), and both of which are in good agreement with 6 nm Ag nanoparticles supported on carbon (368.6 and 374.6 eV) [24], indicating the addition of Sn did not arouse obvious chemical shift of Ag. These results are consistent with Lee's work [25] and confirm the existence of Ag and Sn species on the surface of  $\text{Ag}_4\text{Sn/C}$ .

Fig. 4 presents the cyclic voltammogram (CV) of  $\text{Ag/C}$  and  $\text{Ag}_4\text{Sn/C}$  in  $\text{N}_2$  saturated 0.1 M KOH electrolyte. Three anodic peaks ( $A_1$ ,  $A_2$ , and  $A_3$ ) and one cathodic peak are observed, and identified as described in the following sequence: the small peak  $A_1$  was assigned to the simultaneous formation of a few monolayers of  $\text{AgOH}$  and soluble  $\text{Ag(I)}$  species, peak  $A_2$  was associated

with the formation of inner hydrous oxide layer, peak  $A_3$  was related to the formation of more compact outer oxide layer, and the large reduction peak  $C_1$  could be attributed to the electro-reduction of the complex layer which formed during anodic reaction [26–28]. According to previous reports, the redox of  $\text{SnO}_2$  or  $\text{SnO}$  does not occur within the same potential region [29], therefore it cannot be observed in the cyclic voltammogram in Fig. 3. It can be also seen that the oxidation peaks of  $\text{Ag}_4\text{Sn/C}$  have decreased, and especially, the oxidation process from Ag to  $\text{AgOH}$  or  $\text{Ag}_2\text{O}$  was inhibited so that peak  $A_1$  was almost invisible. These results suggest that less Ag oxide species are formed during the anodic scanning. Moreover, the onset potential of  $C_1$  for  $\text{Ag}_4\text{Sn/C}$  shifted to negative potential, indicating slightly more stable oxide formation. As to the reduction peak of silver oxide for  $\text{Ag}_4\text{Sn/C}$  shifted to positive potential, which seems to be contradiction with the negative shift of onset potential. We think it may result from the deviation of active material mass [30,31].

The kinetics of the ORR for  $\text{Ag/C}$  and  $\text{Ag}_4\text{Sn/C}$  catalyst were studied by the RDE method. Fig. 5a shows the ORR polarization curves of  $\text{Ag}_4\text{Sn/C}$  at different rotation rates. The Koutecky–Levich equation [27] was used to determine the electron transfer number of the catalyzed ORR.

$$1/J = 1/J_k + 1/J_L = 1/J_k + 1/(0.62nFD_0^{2/3}\nu^{-1/6}C_0\omega^{1/2})$$

where  $J$  is the measured current density,  $J_k$  and  $J_L$  are the kinetic and limiting current density, respectively,  $n$  is the number of electron transferred,  $F$  is the Faraday constant ( $96,485 \text{ C mol}^{-1}$ ),  $C_0$  is the saturated concentration of  $\text{O}_2$  in 0.1 M KOH ( $1.2 \times 10^{-3} \text{ mol L}^{-1}$ ),  $D_0$



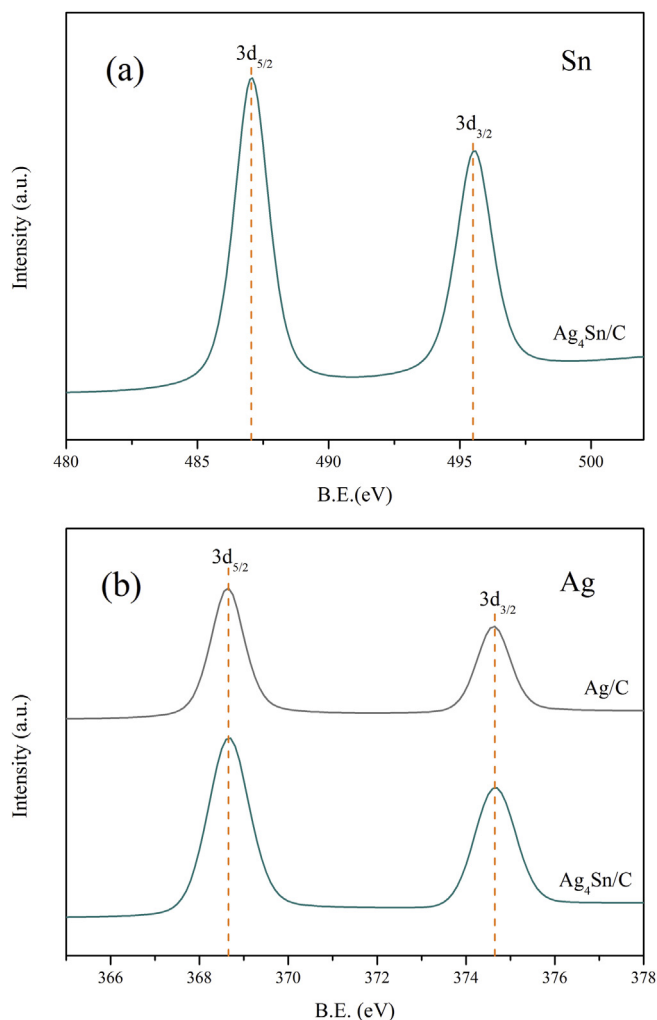


Fig. 3. XPS spectra of (a) Sn 3d in  $\text{Ag}_4\text{Sn}/\text{C}$ , (b) Ag 3d in  $\text{Ag}_4\text{Sn}/\text{C}$  and  $\text{Ag}/\text{C}$ .

is the diffusion coefficient of  $\text{O}_2$  ( $1.9 \times 10^{-5} \text{ cm}^2 \text{ s}^{-1}$ ),  $\nu$  is the kinetic viscosity of 0.1 M KOH ( $0.01 \text{ cm}^2 \text{ s}^{-1}$ ), and  $\omega$  is the electrode rotation rate. Fig. 5b is the K–L plots of  $\text{Ag}_4\text{Sn}/\text{C}$  at potential range from 0.51 V to 0.65 V, while the one for  $\text{Ag}/\text{C}$  is not shown here.

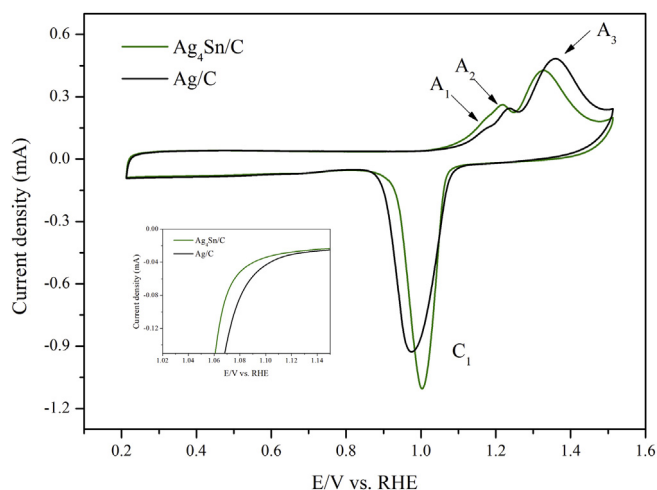


Fig. 4. Cyclic voltammograms of  $\text{Ag}/\text{C}$  and  $\text{Ag}_4\text{Sn}/\text{C}$  in  $\text{N}_2$  saturated 0.1 M KOH solution at a scan rate of  $50 \text{ mV s}^{-1}$ , the inset is the magnification of peak  $\text{C}_1$  onset.

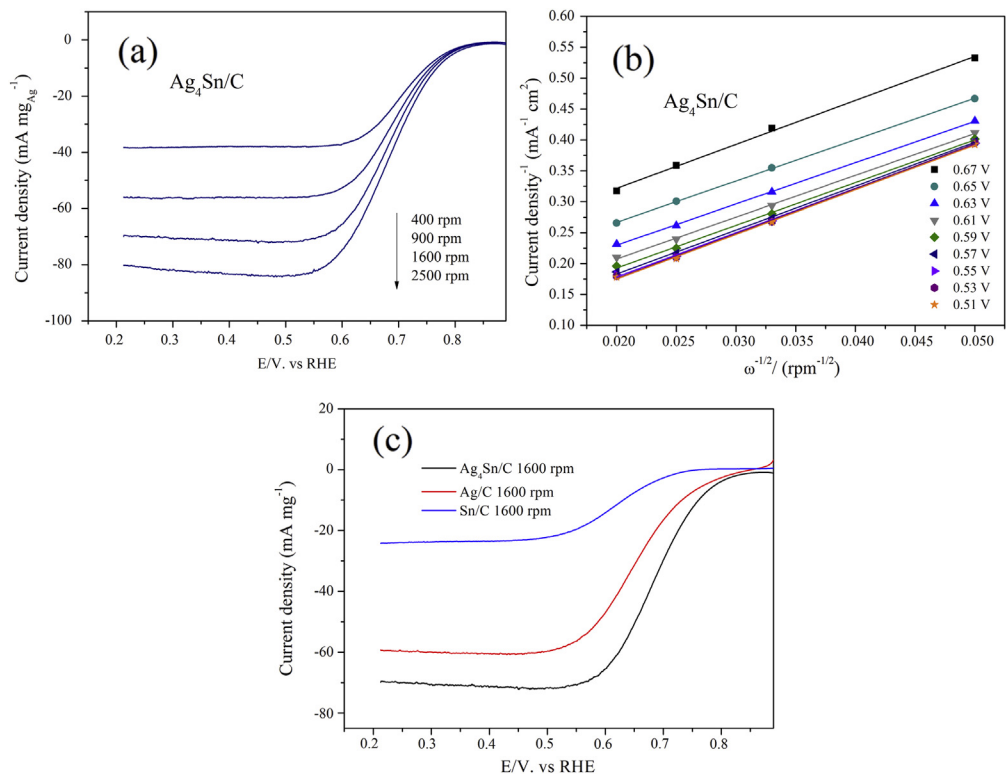
According to the equation,  $n$  could be obtained by intercept of the Levich plots. And  $J_k$  is calculated on the basis of current density at 1600 rpm. It can be seen from Fig. 5b that the slopes of K–L plots for  $\text{Ag}_4\text{Sn}/\text{C}$  remain approximately constant over the potential range, which suggests a consistent electron transfer of ORR at different potentials. Additionally, the calculated numbers of electron transferred on both  $\text{Ag}/\text{C}$  and  $\text{Ag}_4\text{Sn}/\text{C}$  were close to 4, indicating a complete reduction of  $\text{O}_2$  to  $\text{OH}^-$  on both catalysts. Meanwhile,  $\text{Sn}/\text{C}$  shows a very slow kinetic two-electron pathway of incomplete reducing  $\text{O}_2$  to  $\text{H}_2\text{O}_2$ , as shown in Table 1.

Fig. 5c shows the ORR polarization curves for oxygen reduction on  $\text{Ag}/\text{C}$ ,  $\text{Sn}/\text{C}$  and  $\text{Ag}_4\text{Sn}/\text{C}$  catalysts. The results were summarized in Table 1. The onset potential ( $E_{\text{onset}}$ ) shifted to higher value for  $\text{Ag}_4\text{Sn}/\text{C}$  compare to  $\text{Ag}/\text{C}$ , and the kinetic mass activity at different potentials on  $\text{Ag}_4\text{Sn}/\text{C}$  was more than three times larger than  $\text{Ag}/\text{C}$  at 0.6 V, suggesting markedly improved dynamics for ORR relative to  $\text{Ag}/\text{C}$  catalyst. It is worth mentioning that kinetic current density and onset potential are two essential parameters that represent the kinetic and thermodynamic characteristics of the catalytic reaction. Compared with previous reports about Ag-based materials in fuel cell applications,  $\text{Ag}_4\text{Sn}/\text{C}$  shows a comparable oxygen reduction onset potential with silver-molybdate [22], Ag NPs@POM-CNTs [32] and silver–cobalt bimetallic [11] and much higher activity than  $\text{AgMn}$  [33] and  $\text{AgCu}$  [34]. Furthermore, the kinetic current density normalized by geometric area of the working electrode were also given in this work. The calculated results for kinetic current of  $\text{Ag}_4\text{Sn}/\text{C}$  were much higher than previous report of  $\text{AgAu}$  bimetallic [35], yet still lower than  $\text{AgCo}$  [11] bimetallic and  $\text{CoPcF}_{16}/\text{Ag}/\text{C}$  [36] composite which might due to the difference in catalyst loading, morphologies, and their use of gas diffusion electrode.

As to the origin of enhanced oxygen reduction activity of  $\text{Ag}_4\text{Sn}/\text{C}$ , it may be relevant to the surface species of  $\text{Ag}_4\text{Sn}/\text{C}$  nanoparticles, which are Ag and SnO (or  $\text{SnO}_2$ ) confirmed by XPS analysis. Karen E. Swider-Lyons [29] reported higher onset potential of ORR for  $\text{Au-SnO}_x/\text{VC}$  in comparison with  $\text{Au}/\text{VC}$  electrocatalyst and referred this improvement to a bifunctional mechanism, in which  $\text{O}_2$  is chemisorbed onto SnO surface in the form of  $\text{O}_2^-$ , and then further reduced on Au. In addition, this hypothesis was also used to explain the enhanced catalytic activity of  $\text{SnO}_2$  supported Au [37]. Therefore, based on previous researches, such bifunctional mechanism could also be the origin of the enhanced activity of  $\text{Ag}_4\text{Sn}/\text{C}$ . The ORR process could be promoted with the existence of SnO since both Ag and Au are thermodynamically unfavorable to adsorb and split oxygen [38].

Methanol tolerance is a key characteristic for cathode materials in DMAFC applications, therefore it is necessary to examine the effect of methanol in ongoing ORR process. To examine the crossover effect, chronoamperometry measurement was employed for  $\text{Ag}_4\text{Sn}/\text{C}$ , and for comparison,  $\text{Pt}/\text{C}$  has also been tested by the same measurement, the results are shown in Fig. 6. Two arrows in both  $i-t$  curves imply the injection of 2 ml methanol into 100 ml  $\text{O}_2$  saturated 0.1 M KOH electrolyte (i.e. the concentration of methanol was 0.5 M). For  $\text{Ag}_4\text{Sn}/\text{C}$ , the injection of methanol did not arouse an obvious oxidation curve, which indicates that  $\text{Ag}_4\text{Sn}/\text{C}$  is high tolerant towards methanol. Meanwhile,  $i-t$  curve of  $\text{Pt}/\text{C}$  showed a high responsive current corresponding to methanol oxidation, the current density was almost half of the beginning after the curve was stable again, indicating that methanol has a mixed potential effect on  $\text{Pt}/\text{C}$  catalysts [39].

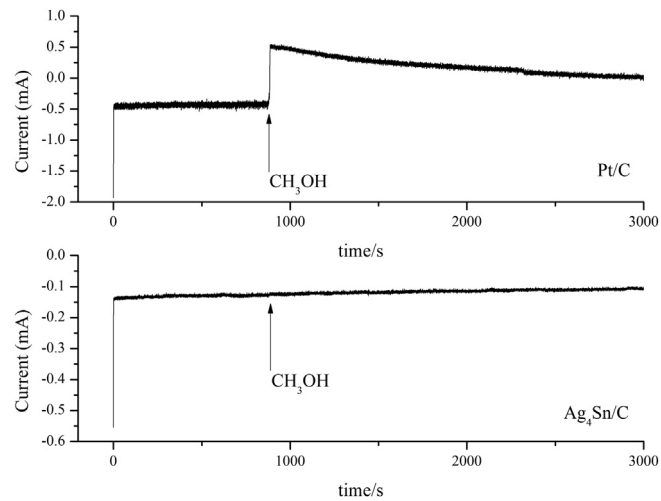
Fig. 7 compares the chronoamperometry curves of  $\text{Ag}/\text{C}$  and  $\text{Ag}_4\text{Sn}/\text{C}$ . After operating the cell at 0.713 V for 24 h, relative current of  $\text{Ag}_4\text{Sn}/\text{C}$  drop to 88%, while  $\text{Ag}/\text{C}$  exhibits more obvious decrease with a current loss of approximately 73%, suggesting enhanced stability of  $\text{Ag}_4\text{Sn}/\text{C}$  intermetallic catalyst.



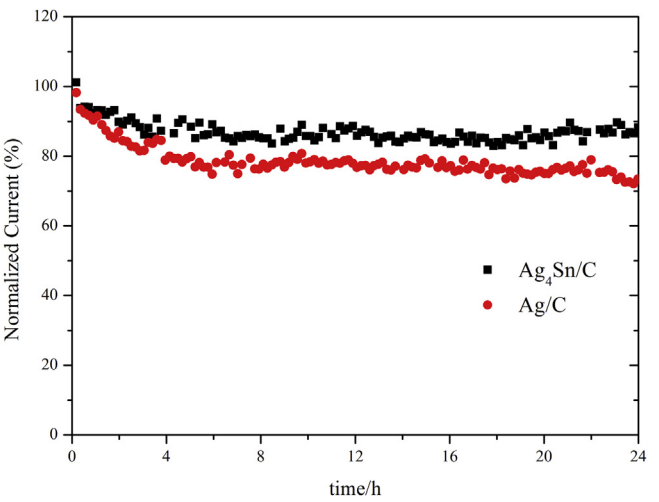
**Fig. 5.** (a) Polarization curves for the ORR in O<sub>2</sub> saturated 0.1 M KOH at different rotation rates on Ag<sub>4</sub>Sn/C at a scan rate of 10 mV s<sup>-1</sup>, (b) Koutecky–Levich plots for Ag<sub>4</sub>Sn/C catalyst obtained at different potentials, (c) ORR polarization curves obtained on Ag<sub>4</sub>Sn/C, Ag/C and Sn/C in 0.1 M KOH saturated with oxygen at 1600 rpm, scan rate: 10 mV s<sup>-1</sup>.

**Table 1**  
Parameters of ORR on Ag<sub>4</sub>Sn/C, Ag/C and Sn/C catalysts.

Catalysts	<i>E</i> <sub>onset</sub> vs. RHE	Kinetic current density (mA mg <sup>-1</sup> )/(mA cm <sub>geo</sub> <sup>-2</sup> )				Number of electron transferred
		0.75 V	0.70 V	0.65 V	0.60 V	
Ag <sub>4</sub> Sn/C	0.77	15.68/1.05	52.01/3.48	172.09/11.50	759.31/50.76	3.9
Ag/C	0.74	8.40/0.53	22.75/1.45	65.13/4.14	209.62/13.34	3.8
Sn/C	0.73	0.25/0.02	3.06/0.26	11.11/0.96	35.70/3.06	2.2



**Fig. 6.** Chronoamperometry curves of Pt/C and Ag<sub>4</sub>Sn/C in O<sub>2</sub> saturated 0.1 M KOH under magnetic stirring (300 rpm), the applied potential was 0.713 V. Arrows indicate the addition of 0.5 M methanol into electrolyte.



**Fig. 7.** Chronoamperometry curves of Ag/C and Ag<sub>4</sub>Sn/C catalysts at electrode potential of 0.713 V in O<sub>2</sub> saturated 0.1 M KOH solution.

#### 4. Conclusions

In summary, a novel Ag<sub>4</sub>Sn/C intermetallic catalyst was synthesized by low temperature chemical reduction method assisted with ultrasonic. This method efficiently synchronized the reduction rate of different metal ionic, therefore ultra-fine and high dispersed intermetallic nanoparticles could be obtained by one step procedure. For the first time, nanoscale Ag<sub>4</sub>Sn/C was used to catalyze ORR in DMAFC application, and showed enhanced catalytic activity. Furthermore, Ag<sub>4</sub>Sn/C also showed high methanol tolerance and good long-term stability in alkaline media. Therefore, the low cost Ag<sub>4</sub>Sn/C with facile synthesis route could be regarded as high potential catalyst for the ORR in DMAFC application.

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